## A New Chlorooxocuprate, $Ca_3Cu_2O_4Cl_2$ , with an Oxygen Defect Intergrowth

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A new chlorooxocuprate,  $Ca_3Cu_2O_4Cl_2$ , is prepared and the crystal structure is determined by Rietveld analysis of X-ray powder diffraction data. This compound crystallizes in tetragonal symmetry with a space group of I4/mmm and with lattice parameters of a = 3.863 Å and c = 21.364 Å. The crystal structure is similar to that of  $Sr_2Ti_3O_7$  except for oxygen vacancies located between two-dimensional Cu–O planes, and for chloride anions occupying apical sites of doubled copper octahedra. Interatomic distances of the compounds are comparable to those of  $Ca_2CuO_2Cl_2$  with a  $K_2NiF_4$  structure. © 1990 Academic Press, Inc.

Recently, the crystal structures and electric properties of layered oxocuprates have been researched intensively because of their potential to become superconductors. well known layered oxocuprate, Α  $La_{2-x}Ba_{x}CuO_{4}$ , was first discovered to be a high- $T_c$  superconductor (1). It has a K<sub>2</sub>NiF<sub>4</sub> structure with two-dimensional Cu-O planes. Another oxocuprate,  $La_2SrCu_2O_{6+y}$ (2), is also known to have a doubleperovskite structure, similar to that of Sr<sub>2</sub> Ti<sub>3</sub>O<sub>7</sub>. In this oxocuprate, oxygen vacancies are located between two-dimensional Cu-O planes, and it has a metallic conductivity when oxidized to y = 0.2 (2), but exhibits no superconductive properties down to 5 K (3, 4). These two types of oxocuprates compose the Ruddlesden-Popper series,  $AO(ABO_3)_n$ , with n = 1 and n = 2(5).

Layered oxocuprates are also seen in halooxocuprates,  $M_2CuO_2X_2$  (M = Ca or Sr, and X = Cl or Br) (6), with a K<sub>2</sub>NiF<sub>4</sub> structure. Although these halooxocuprates have two-dimensional Cu-O planes, as commonly observed in high- $T_c$  superconductors, two halogen anions, instead of oxygen anions, coordinate to a copper atom in *trans*-positions (6). No halooxocuprates with a double-perovskite structure are yet known. During our research on electric properties of Ca<sub>2</sub>CuO<sub>2</sub>Cl<sub>2</sub>, a new double-

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FIG. 1. Idealized crystal structure of Ca<sub>3</sub>Cu<sub>2</sub>O<sub>4</sub>Cl<sub>2</sub>.

perovskite compound, Ca<sub>3</sub>Cu<sub>2</sub>O<sub>4</sub>Cl<sub>2</sub> (Fig. 1), was unexpectedly found. This is the first discovery of halooxometallates with a double-perovskite structure. The new compound, together with  $M_2$ CuO<sub>2</sub> $X_2$ , is thought to be a member of the modified Ruddlesden-Popper series,  $AX(ABO_2X)(ABO_2)_{n-1}$ . We report here the preparation and the structural investigation of the new compound.

This compound was first observed in an X-ray diffraction (XRD) pattern as an impurity phase of a chloride-deficient sample,  $Ca_2CuO_{2+x/2}Cl_{2-x}$ . The almost single phase could be obtained with no detectable Ca<sub>2</sub>  $CuO_2Cl_2$ , after optimizing the starting composition to Ca: Cu: Cl = 3:2:2. The new compound was prepared according to the following procedure. Appropriate amounts of CaCO<sub>3</sub>, CaCl<sub>2</sub>, and CuO were mixed according to a nominal composition of Ca<sub>3</sub>  $Cu_2O_4Cl_2$ . The mixture was ground in agate mortar in a dry box, and calcined at 800°C for 10 hr in flowing  $O_2$ . The products were pulverized, pressed into pellets, and then sintered at 800°C for 10 hr. The crystal structure was investigated by powder X-ray diffraction. The data were taken at room temperature on a Philips 1050/1710 XRD system equipped with a curved graphite monocrometer in the scattered beam path. The intensity data were collected with  $CuK\alpha$  radiation at a 0.02° step width for 10 sec over a  $2\theta$  range from 5 to 90°. A diffraction peak appearing at  $2\theta = 8.3°$  (*d*-spacing = 10.6 Å) suggests the existence of longrange ordering in this compound. The XRD pattern obtained in  $2\theta$  regions from 20 to 90° is shown in Fig. 2. All diffraction peaks, except for a few originating from impurities, could be successfully indexed to a tetragonal unit cell with a = 3.863 Å and c =21.364 Å.

The specific gravity, measured pycnometrically with toluene at 25°C, was 4.09. This value is comparable to the calculated value of 3.98, obtained using a chemical formula of Ca<sub>3</sub>Cu<sub>2</sub>O<sub>4</sub>Cl<sub>2</sub> with the abovementioned lattice constants. The difference between the observed and calculated values may be caused by the impurities and/or hygroscopic nature of this compound.

The lattice dimension of the *a* axis is very close to that of the parent compound,  $Ca_2$   $CuO_2Cl_2$  (3.866 Å) (6), suggesting that the



FIG. 2. Rietveld refinement patterns for Ca<sub>3</sub>Cu<sub>2</sub>O<sub>4</sub> Cl<sub>2</sub>. The solid lines are calculated intensities, crosses overlying them are observed intensities, and  $\Delta y_i$  is the difference between observed and calculated intensities. The short vertical lines mark the positions of calculated peaks.

a-b plane of the new compound is constructed from two-dimensional Cu-O planes. On the other hand, the lattice dimension of the c axis is 1.4 times longer than that of Ca<sub>2</sub>CuO<sub>2</sub>Cl<sub>2</sub>. This value is comparable to the 1.47 obtained as a ratio of the *c*-axis lattice dimension of  $La_{1,9}Ca_{1,1}Cu_2O_6$ (19.420 Å) (7) to that of La<sub>2</sub>CuO<sub>4</sub> (13.17 Å) (8). These similarities as well as the reflection condition with h + k + l = 2n strongly suggest a structural model of Ca<sub>3</sub>Cu<sub>2</sub>O<sub>4</sub>Cl<sub>2</sub> (Fig. 1), which is a double-perovskite structure closely related to that of  $La_2SrCu_2O_6$ , where three  $Ca^{2+}$  cations occupy  $La^{3+}$  and Sr<sup>2+</sup> sites and two Cl<sup>-</sup> anions are located on apical sites of doubled Cu octahedra.

On the basis of the above structural model, the crystal structure of the present compound was refined by Rietveld analysis of X-ray powder diffraction data with the RIETAN program (9). Atomic scattering factors used were those of  $Ca^{2+}$ ,  $Cu^{2+}$ ,  $O^{2-}$ , and  $Cl^{-}$  ions, and anomalous dispersion was corrected. Intensity data in  $2\theta$  regions from 20.0 to 90.0° were used for the refinement, except for regions of  $28.1-28.4^{\circ}$ ,  $35.9-36.2^{\circ}$ , and  $43.5-44.0^{\circ}$ , which contain very weak peaks due to impurities. Preferred orientation was not corrected.

Final crystal-structural parameters are listed in Table I. Quite low *R* factors were obtained:  $R_{wp} = 3.2\%$ ,  $R_p = 2.4\%$ , and  $R_F = 2.7\%$ . Figure 2 illustrates the profile fit

 TABLE I

 Crystal-Structural Parameters of Ca3Cu2O4Cl2

Atom	Site	x	у	z	<b>B</b> (Å <sup>2</sup> )	g
Ca(1)	2a	0.0	0.0	0.0	0.3	1.0
Ca(2)	4e	0.0	0.0	0.1478	0.6	1.0
Cu	4e	0.0	0.0	0.4265	0.8	1.0
O(1)	8g	0.0	0.5	0.0734	0.9	1.0
O(2)	2b	0.0	0.0	0.5		0.0
CÌ	4e	0.0	0.0	0.2980	0.4	1.0

Note. B is the isotropic thermal parameter, and g the occupation factor. a = 3.863 Å; c = 21.364 Å; I4/mmm.

TABLE	Π
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Selected Interatomic Distances in  $Ca_3Cu_2O_4Cl_2$ ,  $Ca_2CuO_2Cl_2$ , and  $La_{1,9}Ca_{1,1}Cu_2O_6$  (in Å)

	Ca <sub>3</sub> Cu <sub>2</sub> O <sub>4</sub> Cl <sub>2</sub>	Ca <sub>2</sub> CuO <sub>2</sub> Cl <sub>2</sub> <sup>a</sup>	$La_{1.9}Ca_{1.1}Cu_2O_6^{t}$
Cu–O <sub>eo</sub> c	1.932	1.93	1.912
Cu-Cl/Oax	2.746	2.73	2.305
Cu–Ca(1)	3.151	_	
Cu-Ca(2)	3.160	3.15	
Cl-Ca(2)ax <sup>c</sup>	3.207	3.19	_
Cl-Ca(2)eg <sup>c</sup>	2.967	2.98	-
CI-CI	3.415	3.39	
Cu–Cu <sup>d</sup>	3.142	_	3.306
$O-O^d$	3.138	—	3.191

<sup>a</sup> From Ref. (6).

<sup>b</sup> From Ref. (7).

<sup>c</sup> ax: axial (c) direction. eq: equatorial (a-b) direction.

<sup>d</sup> Distance through a  $z = \frac{1}{2}$  plane.

and difference patterns for  $Ca_3Cu_2O_4Cl_2$ , showing that the agreement between the calculated and observed intensities is very satisfactory.

The occupation factor of Cl<sup>-</sup> was refined to 1, which means that no chloride anions are replaced by oxygen anions or vacancies. It was first assumed that an oxygen atom is located at the 2b site. However, refinement of the occupation factor of this site shows that it is vacant.

Table II summarizes selected interatomic distances of Ca<sub>3</sub>Cu<sub>2</sub>O<sub>4</sub>Cl<sub>2</sub> together with those of Ca<sub>2</sub>CuO<sub>2</sub>Cl<sub>2</sub> (6) and La<sub>1.9</sub>Ca<sub>1.1</sub> Cu<sub>2</sub>O<sub>6</sub> (7). The corresponding values of these chlorooxocuprates are very close. In double-perovskite compounds, the difference between Cu-Cu and O-O distances through a  $z = \frac{1}{2}$  plane indicates a puckering of the Cu-O plane. The value in Ca<sub>3</sub>Cu<sub>2</sub> O<sub>4</sub>Cl<sub>2</sub> is small (0.004 Å) compared to that in La<sub>1.9</sub>Ca<sub>1.1</sub>Cu<sub>2</sub>O<sub>6</sub> (0.115 Å), showing that the Cu-O plane in the former compound is almost flat.

The vacancy at the 2b site in Ca<sub>3</sub>Cu<sub>2</sub>O<sub>4</sub>Cl<sub>2</sub> presumably results from a relatively short Cu–Cu distance (3.142 Å) through the  $z = \frac{1}{2}$ plane. The corresponding value in La<sub>1.9</sub> Ca<sub>1.1</sub>Cu<sub>2</sub>O<sub>6</sub> is 3.306 Å and this compound also has a vacancy at the 2b site (7). On the other hand, La<sub>2</sub>SrCu<sub>2</sub>O<sub>6+y</sub> (y = 0.0), having a longer Cu–Cu distance of 3.63 Å (10), can accommodate excess oxygen anions into the 2b site until y = 0.2 (2).

The new compound,  $Ca_3Cu_2O_4Cl_2$ , can be regarded as a member of the modified Ruddlesden-Popper series,  $AX(ABO_2X)$  $(ABO_2)_{n-1}$ , where n = 2. The intergrowth of the oxygen-deficient layer,  $(ABO_2)$ , may produce other long-range-ordered new compounds if possible. The research on electric properties of this series of compounds is now in progress. The strontium analog could not be obtained, but products that partially substituted strontium for calcium were obtained.

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## References

- J. G. BEDNOLZ AND K. A. MÜLLER, Z. Phys. B 64, 189 (1986).
- 2. C. MICHEL AND B. RAVEAU, *Rev. Chim. Miner.* 21, 407 (1984).
- 3. J. B. TORRANCE, Y. TOKURA, A. NAZZAL, AND S. S. P. PARKIN, *Phys. Rev. Lett.* **60**, 542 (1988).
- 4. M. HIRATANI, T. SOWA, Y. TAKEDA, AND K. MIYAUCHI, Solid State Commun., in press.
- 5. S. N. RUDDLESDEN AND P. POPPER, Acta Crystallogr. 11, 54 (1958).
- 6. H. MÜLLER-BUSCHBAUM, Angew. Chem. Int. Ed. Engl. 16, 674 (1977), and references cited therein.
- Y. NAKAI, K. KOKUBU, T. KOUYAMA, H. ASANO, Y. ENDOU, F. IZUMI, AND E. T. MUROMACHI, *in* "Abstracts of the 44th Meeting of the Physical Society of Japan," No. 3, p. 272 (1989). [in Japanese]
- P. J. PICONE, H. P. JENSSEN, AND D. R. GABBE, J. Crystal Growth 85, 576 (1987).
- 9. F. IZUMI, Nippon Kessho Gakkai Shi 27, 23 (1985). [in Japanese]
- 10. N. NGUYEN, L. ER-RAKHO, C. MICHEL, J. CHOISNET, AND B. RAVEAU, *Mat. Res. Bull.* 15, 891 (1980).